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Article 1 **Microfluidics Integration on GaAs acoustic biosensors with a** ² **Leakage-Free PDMS and High Pressure based on Bonding** ³ **Technology** ⁴

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Abstract: Microfluidics integration of acoustic biosensors is an actively developing field. Despite 10 significant progress in "passive" microfluidic technology, integration with microacoustic devices is 11 still in research state. The major challenge remains the bonding of polymers with monocrystalline 12 piezoelectrics to seal the microfluidic biosensors. In this contribution we specifically address the 13 challenge of microfluidics integration with gallium arsenide (GaAs) acoustic biosensors. We have 14 developed a robust plasma-assisted bonding technology allowing strong connection between PDMS 15 microfluidic chip and GaAs/SiO₂ at low temperature (70°C) and low applied pressure. Mechanical 16 and fluidic performances of fabricated device were studied. The bonding surfaces were character- 17 ized by water contact angle measurement, ATR-FTIR, AFM and SEM analysis. The bonding strength 18 was characterized using a tensile machine and pressure/leakage tests. The study showed that the 19 sealed chips were able to achieve a limit- of bonding strength of 2.06 MPa. The adhesion of PDMS 20 to GaAs was significantly improved by use of $SiO₂$ intermediate layer, permitting the bonded chip 21 to withstand at least 8 bar of burst pressure. The developed bonding approach can be a valuable 22 solution for microfluidics integration of several types of MEMS devices. 23

Keywords: microsystems; microfluidics; acoustic biosensor; bonding technology; PDMS- 24 SiO2/GaAs bonding; Leakage test 25

1. Introduction 27

Microfluidics field has emerged as a solution for precise control and manipulation of 28 fluids at a microliter scales [\[1](#page-12-0)–[3\]](#page-12-1). On-chip microfluidics integration is one of the most 29 promising development vectors in particular in the field of biosensors [\[4](#page-12-2)[,5\]](#page-12-3). Microfluidics 30 integrated lab-on-a-chip solutions have been widely used in many applications, such as 31 clinical diagnostics on human physiological fluids, cell biology [\[6\]](#page-12-4), detection of tumor 32 cells, biochemical detections, electrophoresis, biochemistry, PCR [\[7\]](#page-12-5), DNA analysis, sin- 33 gle-cell trapping, droplets microfluidics [\[8\]](#page-12-6), biosensors and more [9]. Recently introduced 34 microfluidic biosensors have the advantages of portability, high precision, easy applica- 35 tion, and high-throughput parallel processing [\[10\]](#page-12-7). Ma et al. [\[11\]](#page-12-8) showed that microfluidic 36 channel could lower down the detection limit of endotoxin with the confined space and 37 enhance Van der Waals force. Zhang et al. [\[12\]](#page-12-9) used microfluidic channel with biosensor 38 for detection of Salmonella using Fe-nanocluster amplification and smart phone imaging. 39

Micromachining processes open up the possibility to combine micro-sensors and mi- 40 crofluidics onto a single chip. Various sensing technologies have been integrated in mi- 41 crofluidic systems (e.g. optical, conductive, acoustic, radio frequency and other) making 42 possible to assess physical properties of bio-analytes on a chip level. Ability to complete 43

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The integration of biosensors with microfluidics circuits is in the core of the develop- 46 ment of integrated bio-analytical chips. Among all existing approaches, acoustic biosen- 47 sors become an important tool to study molecular interactions at the surface. The acoustic 48 biosensors have been widely studied in the detection of gases and liquids [\[13](#page-13-0)[,14\]](#page-13-1). There 49 are different types of acoustic biosensor approaches that were developed during the last 50 decades. Bulk acoustic wave devices (BAWs) became one of the most successful ap- 51 proaches in the field. 52

There are several materials that can be used in the fabrication of acoustic wave sen- 53 sors. GaAs has been shown to be very well suited for biosensing application [\[15](#page-13-2)–[17\]](#page-13-3). GaAs 54 is a microtechnical material that combines piezoelectric properties and the possibilities for 55 devices integration and miniaturization. GaAs can be batch micromachined using Induc- 56 tive Coupled Plasma Reactive Ion Etching as well as using low-cost wet chemical etching 57 [\[18](#page-13-4)[,19\]](#page-13-5) . Beside its beneficial microfabrication facilities, GaAs surface can be chemically 58 functionalized with alkanethiols [\[20\]](#page-13-6) silanes and phosphonates [\[21\]](#page-13-7) . The microfluidics 59 integration of GaAs BAWs has a potential to introduce a novel sensing platform. 60

Microfluidics solutions on the other hand are used to be built based on materials such 61 as silicon, glass, PMMA. These materials are commonly used to manufacture fluidic chan- 62 nels, taking advantage of their good mechanical properties and easiness of surface modi- 63 fication to immobilize affinity tags for binding of target molecules on surfaces. Based on 64 PDMS elastomers became attractive alternative materials for microfluidics due to low cost 65 and their remarkable physical and chemical properties, such as wide temperature range, 66 low stiffness, chemical inertness, biocompatibility, rapid prototyping, optical transpar- 67 ency, non-reactivity, high gas permeability. These features make PDMS a potential mate- 68 rial for various applications such as pattern transfer, as well as for fabrication of the com- 69 plex microfluidics systems [\[22](#page-13-8)[,23\]](#page-13-9) . In addition, its low bonding temperature (lower than 70 100°C) makes it an excellent material for bonding elastomer substrates since many elasto- 71 mer substrates cannot withstand a high bonding temperature. The mass of the state of 72

Reproducible bonding/sealing remains one of the highest challenges for reliable ap- 73 plications of microfluidic systems in biosensors. The popular bonding methods such as 74 anodic bonding for Si/Glass microfluidic devices, thermo-compression bonding or chem- 75 ical assisted bonding [\[24](#page-13-10)–[26\]](#page-13-11) encounter various difficulties when applied for piezoelectric 76 substrates. On the other hand, polymers provide the alternative solutions for microfluid- 77 ics packaging. Recently, several new strategies were introduced to improve microfluidics 78 packaging for the integrated sensor solutions. According to the literature, Kersy et al. used 79 the adhesion promotor GE SS4120, while it does not improve adhesion of PDMS to Teflon 80 [\[27\]](#page-13-12). It deceases the adhesion strength between PDMS-PDMS. However, this method im-
81 proves adhesion of PDMS to silicon, glass and aluminum, only allows the formation of a 82 strong bond between the substrate and an un-structured layer of PDMS. Carlos Luis et al. 83 [\[28\]](#page-13-13) have proposed the use of narrow electrode connectors for minimizing the solution 84 leakage in the PDMS-Au interface. Yong et al. [\[29\]](#page-13-14) have used thermo-compression and 85 laser bonding to fabricate multi-layer glass microfluidic chips. Application of sticky elas- 86 tomer was introduced for epidermal electronics [\[30](#page-13-15)–[32\]](#page-13-16) . Heterogeneous crosslinking of 87 PDMS was applied to enhance adhesion of PDMS to several substrates as seen in Jeong et 88 al. [\[33\]](#page-13-17) . Plasma-assisted bonding was used by Xi et al. [\[34\]](#page-14-0) for improve the bonding be- 89 tween PDMS-coated glass cover plate and silicon substrate. 90

In the current contribution, we develop the solution for microfluidics integration of 91 GaAs biosensors. In particular, we study PDMS-GaAs system where the challenge raises 92 due to GaAs being inert to plasma bonding. To address the challenge, some authors have 93 attempted to increase the adhesion between the PDMS and the substrate by using a gold 94 layer. However, this method is not suitable to seal patterned PDMS with micromachined 95 GaAs. Others have proposed 3-aminopropyl)trimethoxysilane (APTES) and achieved a 96 bonding strength of 406 kPa for PMMA/PDMS bonding [\[35](#page-14-1)[,36\]](#page-14-2). A thermal bonding 97 method is used to bond four-layer microfluidic chip [\[37\]](#page-14-3). In the other approach, PDMS is 98 mixed with a small amount of polyethylenimine solution to prepare a sticky thin layer, 99 which works like a sticky tape to adhere on glass, PMMA, and metal by contact press [\[38\]](#page-14-4). 100 Lastly, Anil et al, [\[39\]](#page-14-5) have developed microfluidics-integrated microscale that comprise 101 an isoporous nanostructured membrane. The GaAs on Ge/Si substrate was first flipped 102 with the GaAs nanopyramids side bonded to a Polydimethylsiloxane (PDMS) substrate, 103 whereas a transparent flexible polymer film was weakly bonded by Van der Waals force 104 [\[40](#page-14-6)[,41\]](#page-14-7). The SiO₂ layer was used to increase PMMA bonding capability to PDMS in fabri- 105 cating gas micro valves. Ahmad et al, have showed strong and irreversible bond of PDMS 106 on PMMA when it is covered with $SiO₂$ nano particles [\[42\]](#page-14-8). 107

In this study, we developed the approach for irreversible and leakage-free plasma 108 assisted bonding to integrate PDMS microfluidic channel with GaAs substrate. For this 109 purpose, we combine thin-film SiO₂ intermediate layer on GaAs substrate with plasma $O₂$ 110 treatment and low-temperature annealing. In brief, this bonding technology is obtained 111 in four main steps: 1) SiO₂ deposition, 2) plasma O₂ treatment, 3) chip alignment and bond- 112 ing, and 4) annealing at low temperature 70°C. This method is appealing for its compati- 113 bility to traditional replication method using PDMS and the surface structures can be re- 114 tained. The characterization of PDMS and $GaAs/SiO₂$ surfaces before bonding was veri- 115 fied by contact angle, Attenuated Total Reflectance-Fourier Transformed Infrared Spec- 116 troscopy (ATR-FTIR), AFM and the SEM analysis. Bonded chips were characterized using 117 a tensile machine strength bonding equipment on PDMS and a leakage bench test. 118

2. Materials and Methods 119

2.1 Materials 120

Polydimethylsiloxane (PDMS, Sylgard 184) was obtained from Dow Corning Toray 121 Corp. SU-8 3025 photoresist was purchased from MicroChem (Newton, MA, USA). Un- 122 doped, 3-inch in diameter and 625 ±25 µm thick double-side polished GaAs (100) ± $0.5°$ 123 wafers (AXT, Inc., Fremont, CA, USA) were used to fabricate the biointerface chips. Ace- 124 tone (ACP Chemicals, Saint-Léonard, QC, Canada) were used to clean substrates. Red 125 color dye was purchased from Shanghai Macklin Biochemical Co., Ltd. (Shanghai, China). 126 Syringe pump LSP02-1B (Longer Pump, China) was applied to generate constant flow rate 127 in microfuidic chip. Diamond saw dicing was used to cut PDMS for placing microfuidic 128 chip. SEM microscope (Thermofisher APREO S Low-vacuum SEM and 30 mm² SDD EDX) 129 was used to observe in detail the microfuidic channel. AFM scans of different dimensions 130 were recorded in order to have a representative sampling of the surface roughness of 131 GaAs/SiO2. The AFM cantilever had a nominal resonance frequency of 330 kHz, a force 132 constant of 42 N/m, a length of 125 μ m, and a mean width of 30 μ m. 133

2.2 Methods and equipment 134

The fabrication process of the PDMS microfluidic channel is schematically illustrated 135 in figure 1. SU-8 mold of 70µm in thickness was fabricated with SU-8 3050 onto a 1mm- 136 thick silicon wafer using standard photolithography processes including spin coating, 137 pre-baking, exposure, post-baking and development. The SU-8 mold was then used to 138 replicate PDMS microfluidic channel. PDMS with thickness of 3 mm was prepared by 139 pouring the mixture of low-modulus PDMS (component ratio A:B = 1:10, Sylgard 184, 140 Dow Corning). 141

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145 **Figur[e 1](#page-2-0)**. Fabrication process of the PDMS microfluidic channel by replica moulding: (1) fabrication of SU8 master mold using photolithography; (2) Pouring of the mixture of PDMS prepolymer and 146 curing agent into the master mould and allowing it to solidify; (3), (4) Peeling of the solidified PDMS 147 from the master mould and cutting; (5) Punching the inlet and outlet holes; (6) Microfluidic channel. 148

Air bubbles that appeared during the mixing were removed using a vacuum desica- 150 tor, followed by baking at 80 °C for 2 h, followed by pouring onto the SU-8 mold to have 151 a sticky layer of about 300 μm. Finally, the prepared PDMS structures were peeled off 152 from the mold and small inlet and outlet holes were punched. Images of the SU-8 mold 153 and the fabricated PDMS microfluidic channel are shown in figure 1. The resulting 154 channels have a hight of 60 μ m and a width of 300 μ m. The GaAS surface was covered 155 with the SiO² layer by RF reactive magnetron sputtering MP450S machine (Plassys, 156 France). Plasma activation process was performed in 15mTorr, parameters like power 150 157 W, oxygen flow 80 sccm. 158

Figure 2 presents an experimental setup to measure flow rate and pressure in the 160 microfluidic channel. Once the devices were fabricated and assembled, we connected 25 161 mL syringes to Tygon (Sigma-Aldrich) tubing to control the volume of air pumped in and 162 out the control channels. A syringe pump is mounted to accurately displace the syringe 163 plunger. Additionally, we connected the inlet of the microfluidic device with Tygon tub- 164 ing to a supplementary syringe pump to control the flow rate of our medium and cell 165 sample. In our experiments, the syringe volume varies from 0 mL to 25 mL. 166

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Figur[e 2](#page-3-0). Experimental setup to control flow rate and pressure in the microfluidic channel.

Hence, the pressure difference varies from 0 to 8 bar. With the syringe pumps, the 170 single-layer valves in this device can be accurately controlled without using more compli- 171 cated electropneumatic systems. An LG16 (Sensirion) microfuidic control system was 172 used to deliver fluid to the channel of the test device and to monitor the applied pressure. 173 Images of the microfluidic channel were obtained with optical microscope (Mitutoyo 174 FS70, Mitutoyo Corp., Kawasaki, Japan) and a camera (IDS µEye, IDS Imaging Develop- 175 ment Systems, Obersulm, Germany) with a spatial resolution of 5.5 μ m/pixel. During our 176 experiments, a flow sensor connected to a PC is placed for continuous recording of the 177 flow rate and pressure flow in the microfluidic channel. 178

3. Results 179

3.1. SEM Characterization 180

The test PDMS microfluidic structure about 3mm in thickness was cut along the chan- 181 nel length by a sharp blade. Resulted PDMS membrane with exposed microchannel (see 182 figure (3A) was coated with Cr thin layer to be analyzed in the SEM microscope. As shown 183 in figure (3B), the shape of a 400 μ m-wide microchannel pattern casted in PDMS in a single 184 millimeter scale is well preserved. The same state of the state of

To show that presented bonding method can preserve the channel profile as pure 186 PDMS does, we bonded a PDMS microfluidic structure to a 650 μm thick, (100) oriented 187 GaAs substrate, previously covered with 100nm $SiO₂$ layer. Figure (3C) shows the cross- 188 sectional profile of the 80 μ m-high microchannel after bonding, proving that our way of 189 chip packaging with sandwich structure is safe. The two subtrates (GaAs and PDMS) are 190 covalent bonded after plasma oxygen treatment of SiO² intermediate thin layer and 191 thermal annealing. 192

Microfluidic channel

Figur[e 3](#page-4-0). Characterization of microfluidic channels, by SEM: A) General view on PDMS membrane 195 with channel and intlet/outlet holes, (B) microfluidic channel, and (C) cross section of GaAs/SiO₂ 196 bonded to PDMS. The state of the state o

3.2. Activation-Characterization of PDMS and GaAs/SiO² surfaces- interfaces 199

The hydrophobicity of PDMS is associated with the organic methyl groups present 200 in the chemical structure of PDMS. The microchannel was cut out of the mold, following 201 by oxygen plasma treatment to render the PDMS surface hydrophilic. We prepared our 202 bonding technology by a combination of surfaces treatment and annealing (figure 4). Ox- 203 ygen plasma treatment shows to be the most rapid process to increase of the hydrophilic- 204 ity of PDMS surface by removing hydrocarbon groups and introducing polar silanol (Si- 205

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OH) groups via oxidation. The activation process duration was 60 seconds. Subsequently, 206 two surfaces were bonded by bringing them into contact under pressure 2N followed by 207 a heat treatment at 70°C for 1 h. 208

Figure 4. (1) and (2) Schematic presentation of GaAs/SiO² and PDMS surfaces modifications by 212 plasma O2 of, (3) Bonding structure and annealing at 70°C. 213

To characterize the surfaces modification of PDMS replica and GaAs/SiO₂ substrate 214 after each step of plasma treatment, the contact angles (CA) measurements were per- 215 formed. Water droplets (5µL) were deposited on the surface of each studied surface. As 216 shown in figure 5, after the surface activation, the contact angle has dropped from $100[°]$ to 217 53.8◦ after the oxygen plasma treatment of PDMS, and from 41,3◦ to 11,9° after the oxygen 218 plasma-treated GaAs/SiO2. The drastic decrease of contact angles indicated that the hy- 219 drophobic surface of PDMS became hydrophilic due to the hydroxyl terminals on the 220 plasma-activated PDMS surface. The surface of PDMS after plasma treatment has low 221 surface energy due to the weak intermolecular forces between the methyl groups and the 222 strong (Si–O) and flexible (Si–O–Si) siloxane chain [\[43\]](#page-14-9). 223

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Figur[e 5.](#page-5-0) Contact angles for water droplet on the different surfaces, (A) PDMS before treatment, (B) 227 PDMS activated with plasma treatment, (C) GaAs/SiO₂ before plasma O₂ and (D) after plasma O₂ 228 treatment 229

The surface functional group of GaAs, silicone dioxide and PDMS were analyzed by 230 using ATR-FTIR in the MIR spectral region from 4000 to 500 cm⁻¹ (λ =2.5-20 μ m) in order to 231

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study the effect of oxygen plasma on surface modification. The IR transmittance spectra 232 are presented in figure 6. The peaks between 2,950 cm⁻¹ and 2,970 cm⁻¹ correspond to the 233 asymmetric Si–OH bonds of PDMS. The peaks at 1,257 cm−1 and 1,010 cm−1 are attributed 234 to CH³ asymmetric deformation and Si–O–Si asymmetric deformation of PDMS respec- 235 tively. A high transmittance on the PDMS substrate in a visible light domain is attained. 236 In comparison, the transmittance of the PDMS substrate is 96% , and that of the GaAs/SiO₂ 237 is 85%, which leads the PDMS transmittance to be 11% better than the GaAs/SiO2. 238

Figur[e 6](#page-6-0). ATR-FTIR spectra recorded from: (a) PDMS polymer before and after plasma treatment, 241 (b) the GaAs/SiO² (100) surface before and after plasma treatment. 242

To ensure the success of surface modification, ATR-FTIR measurement was con- 243 ducted for six different substrates to verify reproducibility with a maximum error of 5%. 244 Due to the wafer thickness of 650 μ m, the GaAs disks support only internal reflections. 245 Especially around 1200 cm−1, silicon oxide possesses a vibrational mode, and thus reduced 246 transparency in the so-called molecular fingerprint region. The presence of peak in 1116 247 $cm⁻¹$ corresponds to a thin layer of Si-O. Beside the transmittance peak in 1018 cm⁻¹ which 248 relates to methyl groups there is trace of chloroform in the silicon dioxide intermediate 249 layer. After the treatment of the oxygen plasma, a large amount of hydroxyl groups are 250 produced in the surfaces of both the silicon dioxide and the PDMS. The later conformal 251 contact of the two surfaces will yield a large amount of Si-O-Si bonds, which will form the 252 strong adhesion between the two surfaces. A reorganization of the short polymeric chains 253 supporting the creation of polar groups can be considered resulting in an increase in tem- 254 perature. One major advantage of the temperature increase is that the polymerization 255 takes place while preserving the functionality of the monomer. Moreover, the increased 256 cross-linking density in the PDMS directly influences the strength of covalent bonding. 257

*3.2. Optimization of the Bonding microfluidic channel with GaAs substrate***.** 258

3.2.1. Test with Plasma O² and annealing 259

The surface of GaAs at different fabrication steps was analyzed with AFM within a 260 scanning area of 3_{km}^{*} 3_{km}. The surface morphology of the silicon dioxide was examined 261 using the AFM, in which the grain sizes can be clearly observed and it is noticed that the 262 surface roughness was 0.696 nm. As it can be seen from the AFM images (shown in figure 263 $7(A)$ and $7(B)$, the grain size is in the range of 20 - 58 nm.

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Figure 7. Surface morphology of SiO₂ thin film deposited on GaAs substrates a) before and b) after 268 O2 plasma treatment (AFM images, 3x3 μm² , contact mode, silicon nitride tips (0.32 N/m), 512*512 269 pixels resolution) c) SEM image of bonding interface (scale 5 μ m). 270

It was found that the average roughness Ra of SiO² surface has decreased from 552 271 pm to 441pm (Table 1) after plasma treatment. It is seen that both the average roughness 272 and root mean square roughness of the PDMS surface are a factor around 465% higher 273 than those of the GaAs/SiO₂. 274

Tabl[e 1.](#page-7-0) The root mean square Roughness Rq and the average roughness Ra values to various sub- 275 strates. (PDMS value from literature [\[44\]](#page-14-10)) 276

Sample	RMS Roughness Rq	Average Roughness Ra	
GaAs/SiO ₂	709 pm	552 pm	
GaAs	114 pm	90 pm	
GaAs/SiO ₂ after plasma O ₂	697 pm	441 pm not reported	
PDMS	1441 pm		
PDMS treated by plasma O ₂	40031 pm	not reported	

 Obtained results indicate that smoother surface of GaAs/SiO² can be achieved by 278 performing the above-mentioned modification. However, it was shown by Zahid et al. 279 $[44]$ that the plasma O₂ treatment of PDMS leads to a significant increase of its surface 280 roughness, from Rq=1441 pm of freshly PDMS to Rq=40031 pm after plasma treatment. 281 The results further confirmed that the PDMS layer adhered on the silicon dioxide wafer 282 during the cast molding process. 283

3.2.2 Bonding strength evaluation 285

The bonding strength was investigated using destructive mechanical tensile test 286 method on square $10x10mm^2$ bonded pair. Beside the GaAs/SiO2-PDMS bonding config- 287 uration, the bonding strength of two other interesting systems, PDMS-Glass and PDMS- 288 LiNbO₃ was evaluated. The PDMS, glass and LiNbO₃ substrates were cut using diamond 289 saw dicing whereas the GaAs/SiO₂ were cut with a cleavage method. The all substrates 290 were cleaned with acetone and isopropanol to remove dust and organic contaminants. 291 The bonding strenght between the different substrates and PDMS was determined using 292 a mechanical tester for micro-components (Nordson DAGE 4000Plus), equipped with a 293 250kg cartridge (50kg range used). The specific horizontal setup of the tensile test and the 294

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assembled sample are shown in figure (8A). Two steel sample holders were used to sand- 295 wich the bonded sample in between. The upper part of PDMS replica and the backside of 296 all substrates were glued on steel sample holders using a single component instant adhe- 297 sive Loctite 480 (Henkel Adhesives). The sandwiched sample and its holders were fixed 298 horizontally in the special stud-pull fixture of the tester that eliminates side forces due to 299 gimbal construction. During the tensile test, the motorized stage moved horizontally at a 300 uniform speed of 10 μ m/s, while the load force and displacement were recorded simulta- 301 neously until failure occurred. Failure mode was determined by optical inspection of frac- 302 ture surface (interface). The tensile strength of the tested sample was calculated by divid- 303 ing the measured maximal load force at the bond failure. 304

All tensil, tests were conducted on the samples that undergone plasma treatment and 305 thermal annealing at 70° C for one hour. The load force curves for all tested configurations 306 are presented in the figure 8(B-left) showing that the bonding strength of GaAs/SiO2- 307 PDMS is consistently higher when compared to other configurations with LiNbO₃ and 308 glass substrates, i.e. 20.34kg, 13.05kg and 7.93kg, respectively. The use of intermediate 309 bonding layer to PDMS substrate increased the bonding strength most significantly: in- 310 crease of 55% compared with glass substrate and of 156 % compared with LiNbO3. The 311 achieved bonding strength of PDMS and $GaAs/SiO₂$ to PDMS is \sim 2.06 MPa. For all sam- 312 ples, the strength response exhibits conventional behavior: increase over time until reach- 313 ing the broken and sharply decreases. The breaking was observed at 12s, 18s and 19s for 314 GaAs/SiO₂, LiNbO₃ and glass respectively. 315

Figure 8. Bonding strength evaluation: (A) Setup of the tensile test: (1) assembly of the bonded sam- 317 ple: the sample is fixed on two steel sample holders, die is places inside the cap rod is screwed into 318 the block and the die is glued between support plate and block, (2) position of the tool behind stud 319 pin (3) the sample mounting for setup of the tensile test. (B-left) Load-time curves obtained for all 320

substrates bonded to PDMS, (B-right) reproducibility histogram for bonding strength. (C) Photos of 321 broken interface for the PDMS-GaAs/SiO₂, PDMS-Glass and PDMS- LiNbO₃. 322

Figure 8 (B-right), shows the average bonding strength for different samples bonded 323 by the presented process. A good reproducibility for our measurements was observed, 324 with a maximum error of 6% caused by the dispersive imprecision on the dimensions of 325 the bonding area. Figure 8 (C) , shows photos of broken structures after the tensile test. In 326 the case of GaAs/SiO₂-PDMS chip, failure was generally observed on the SiO₂-PDMS 327 bonding interface. But in the case of Glass and LiNbO₃ samples, failure was observed in 328 the PDMS volume were in good agreement with literature reports [27] [36] [45]. The result 329 can be explained by the higher bonding strength between PDMS-Glass and PDMS-LiNbO $_3$ 330 than the breaking point of cured PDMS materials. In fact, when two amorphous surfaces 331 are brought into contact at elevated temperatures where the molecular chain mobility is 332 high, adhesion occurs at the interface. The chain ends penetrate to the opposite substrate 333 of the interface in the surface layer, leading to high bonding strength. Also, the increase 334 in the surface roughness enhances the adhesion property of substrates, the surface rough- 335 ness of LiNbO₃ was larger than glass no matter what the plasma activation time was ac- 336 cording the findings of Xu et al $[46]$. To go further in our interpretation, we would like to 337 hypothesize the large mismatch in the coefficient of thermal expansion between LiNbO3 338 and glass (14.4 (x, y-axis)- 7.5 (z-axis) × 10[.]6 K⁻¹ for LiNbO3, and 0.56 × 10[.]6 K⁻¹ for glass) will 339 produce large thermal stress at the bonding interface may lead to has a significant influ- 340 ence on the value of load force between glass end LiNbO₃ [46][47]. 341

Since the SiO₂ layer was also locally detached from the GaAs surface, this failure 343 mode is partially affected by limited adhesion of SiO2 layer on GaAs that indicates that the 344 bonding strength is higher than the coherence of PDMS material. The high stiffness of 345 GaAs promotes interface failure. Moreover, the high value of the Young's modulus of 346 GaAs (118 GPa) implies that the fracture takes place at the interface. 347

3.2.3 Leakage tests 349

 (A)

In order to validate a leakage-free performance of the bonded GaAs/SiO2-PDMS 350 chips, the channel inlet/outlet holes were equipped with epoxy-sealed connectors and the 351 leakage test was done under defined flow conditions figure (9A). The flow rate was in- 352 creased from $10 \mu l/min$ to $4000 \mu l/min$ by an increment of $10 \mu l/min$ every 30s, looking for 353 the maximum working pressure with no leakage appeared. A red dye solution was used 354 for easier optical microscope inspection of potential leakage at the border of the channel. 355

 (B)

Figure 9. (A) Image of single microfluidic channel (height=60µm, witdh 300µm, length 3cm) in- 357 fused with a red dye at 3000 µl/min and 5000 mbar pressure, (B) Increase of working pressure and 358 flow rate on the microchannel. 359

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In the leakage test, the design chips based PDMS-SiO2/GaAs were kept intact (i.e. 360 without dissemination of red dye solution into the bonded interface) until 8 bar of work- 361 ing pressure, maximum available in our experimental setup. It is clearly seen in figure 362 (9A) that the channel is well defined and no leakage is observed at flow rate of 4000µl/min 363 and 8000mbar. The maximum working pressure and the maximum flow rate until the red 364 dye are shown in figure (9B). 365

4. Discussion 366

Due to the challenges related to the use of PDMS as the structural material, most 367 studies have focused on finding alternative materials. Although other polymers, such as 368 PS, PMMA, TPE, COP, and photoresists have been used for fabrication of microfluidic 369 devices [\[45\]](#page-14-11)[48[-50\]](#page-14-12). Many different problems releated to the bonding of the microfluidic 370 channel on piezoelectric substrate like GaAs (100) were identified. Bonding processes are 371 required for assembly of microfluidic devices, made of two or more components. This can 372 be achieved by using of double-sided tape, glues, or solvent bonding [\[51\]](#page-14-13). The bonding of 373 PDMS on different substrates have been reported in the scientific literature [\[36\]](#page-14-2), but the 374 solvents used in the bonding process can also strongly influence the growth of cells cul- 375 tured in the microfluidic devices [\[52\]](#page-14-14). Wu et al. used (3-Mercaptopropyl) trimethoxysilane 376 (MPTMS) which was a chemical coupling reagent to modify the surfaces of the noble met- 377 als and the PDMS to improve their adhesion [\[53\]](#page-15-0). Yong et al. proposed a novel approach 378 to fabricate multi-layer glass microfluidics chips which comprises laser cutting and ther- 379 mocompression bonding [\[29\]](#page-13-14). Among the methods presented above, special treatments 380 processes or the processes of adding additional chemical reagents are required to achieve 381 PDMS adhesion. This work presents an approach to assess how to improve PDMS / GaAs 382 bonding based on the combination of low temperature (<100°C) plasma/thermal treat- 383 ments and the use of appropriate intermediate bonding layer. We proposed a new solu- 384 tion for bonding of PDMS microfluidic cell on the gallium arsenide substrate, covered 385 with silicon dioxide thin intermediate layer. The aim of the characterization step was to 386 optimize the bonding quality of the multilayer GaAs/SiO2-PDMS. The study consisted on 387 experimental investigations. We have characterized the bonding interface by various 388 measurement techniques (MEB, AFM, ATR-FTIR, CA). 389

In order to prove the usefulness of the proposed solution, the elements and the chem- 390 ical bonds on the PDMS surfaces have been determined by ATR-FTIR analysis. The rough- 391 ness and topography of various treated and non-treated PDMS and GaAs/SiO² surfaces 392 were also analyzed using AFM. The GaAs/SiO2-PDMS samples fabricated according the 393 proposed method were able to withstand the load force until 20,34 kg without failure, 394 which corresponds to bonding strength of 2,06 MPa which was the highest value we got. 395 This is substantially higher than the bonding strength of other tested microfluidic systems, 396 such as Glass-PDMS or LiNbO₃-PDMS. According to the literature it is reasonable to con-
397 sider the tensile strength of PDMS which is much higher than this bonding breaking point 398 [\[54\]](#page-15-1). An analysis system was created to measure the bonding strength of the bonded chips. 399 Zhen et al. showed a bonding strength of over 1.4 MPa for PDMS and PMMA $[36]$. Yong 400 et al. reported the optimal pressure 0.4 MPa [\[29\]](#page-13-14). Kersey et al. employed that the adhesion 401 promoter GE SS4120 can improve the adhesion of PDMS to silicon, glass and aluminum 402 substrates, with bonding strength values 0.841 MPa, 0.847 MPa and 0.488 MPa respec- 403 tively [\[27\]](#page-13-12). From our experiment the bonding strength obtained is higher. In addition, in 404 terms of time consumption, our method was at least times faster than other bonding meth- 405 ω ds. ω 406

In the plasma treatment process, there is only a small amount of hydroxyl groups 407 (the inherent hydroxyl groups) on the surfaces of PDMS and silicon dioxide layer. Xiang- 408 dong et al, mentioned that the contact of such two surfaces will produce a small amount 409 of Si-O-Si bonds, which can only form a week adhesion force [\[55\]](#page-15-2). Temperature was used 410 to improve the adhesion performance of the bonding technology. Annealing contributes 411 to the stabilization of the bonding layer, improving the cross-linking density in the PDMS 412

and favorized the orientation molecular chains. For silicon, glass substrates, bonding tem- 413 perature over 100° C, but for polymer substrates, this would greatly affect the bonding 414 performance. Winnie et al. [\[56\]](#page-15-3) have used hot embossing technique for bonding PDMS- 415 PMMA substrate at 90 \degree C for 3 hours. A temperature of 605 \degree C was used to bond five-layers 416 of glass microfluidic devices. An additional annealing at 65° C for 1 hours on the PDMS 417 and PS surfaces improved bonding and allowed stabilization of the higher SEF for a longer 418 period of time ($(>3$ days). In this work, to enhance the bonding, we have combined plasma 419 treatment and annealing. At our experiment after plasma treatment, both parts are aligned 420 and pressed together while completing the curing process at low temperature 70° C for 1 421 hour. Pre-stress uniformly applied during bonding influences significantly the orientation 422 of the polymer chains. Hammami el al. [\[57\]](#page-15-4) showed that the combination of temperature 423 and stretch promotes the orientation of molecular chains in the dielectric elastomer. Sub- 424 sequently the combination of temperature and small deformation promotes bonding of 425 our system. The bonding strength of the adhesive sandwiched between the PDMS and the 426 GaAs/SiO₂ substrates remained the same, even after six months. 427

The chip holder ensures that leakage only occurs at the bounded interface between 429 the PDMS and GaAs/SiO² substrates, i.e., not at the tubing connector. When the leakage 430 was observed, the bonding strength between the PDMS/substrates chip could then be de- 431 termined. When comparing the strength of our method with GaAs/SiO2-PDMS and an- 432 odic bonding [\[58\]](#page-15-5) and the rapid Pyrex glass bonding, the strength of our bonding was 433 higher than that of the anodic bonding, are higher than that of the rapid Pyrex bonding. 434 No leak was observed in the tested GaAs/SiO2-PDMS samples until maximal available 435 working pressure of 8 bar. To our knowledge, it's the highest leakage-free pressure re- 436 ported in the literature for PDMS-based bonding systems. A comparison between previ- 437 ously reported results of leakage test is shown in Table.2. 438

Sample	PDMS [bar]	Method	Ref
GaAs	≥ 8	plasma O ₂ , SiO ₂ , annealed	This work
Glass	5.1	plasma oxygen ICP	[49]
PDMS	6.7	plasma oxygen RIE	[48]
$SU-8$	1.5	plasma oxygen, small amount of PEIE and temperature	[38]
Glass/Au	2.38	plasma oxygen and narrow electrode	[28]
TPE	4.7	plasma oxygen and thermal bonding	[59]
PMMA	2.5	plasma oxygen	[44]

Table 2. Maximum pressures obtained from previously reported methods are also shown 439 for comparison. 440

Overall, this study presents a method for evaluating whether PDMS can be used as 442 a reliable structural material for microfluidic devices in order to enhance the performance 443 of acousto-fluidics-biosensors based on GaAs. Our contribution is discussed only in terms 444 of basic technological challenge to coupled PDMS with GaAs. Nevertheless, presented 445 results characterizations are likely to contribute to the improvement of the performance 446 of the microfluidics systems combining PDMS and GaAs. 447

5. Conclusions 448

We have presented a novel combination between GaAs and PDMS which has would 449 enable the development of an increasingly in-demand array of new applications, includ- 450 ing those requiring high flow rates and high pressures. The combination of $SiO₂$ interme- 451 diate layer plasma oxygen and low-temperature annealing and $SiO₂$ intermediate layer 452

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significantly improves bonding of PDMS to GaAs substrate. In our acoustic biosensor ap- 453 plication, one can assume a maximum pressure driving around 8 bar for the fluid. The 454 bonding area of microfluidic devices can withstand a stress about 2.06 MPa. Additionally, 455 this bonding method does not require wet chemical treatment of bonded surfaces which 456 may be prohibited in some application. Bonding features were evaluated using different 457 methods, bonding strength and leakage test. Compared to the previous studies, our bond- 458 ing method has shown a robust and rapid fabrication technology as well as superior bond- 459 ing strength and leakage-free pressure. The obtained results can be valuable for research 460 and development of integrated microfluidic devices based on PDMS material in general. 461

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